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# Anion-exchange synthesis of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> core/shell composites with enhanced visible and NIR light photocatalytic performance and the photocatalytic mechanisms



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#### ABSTRACT

A series of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites with well-defined core/shell structures were synthesized via an in-situ anion-exchange reaction between Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons and Na<sub>2</sub>S solutions. The obtained samples were characterized by XRD, BET, AFS, SEM, TEM, EDX, XPS, DRS and PL. The Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites exhibited excellent photocatalytic activity towards the degradation of MO under both visible and NIR light irradiation. The optimal composite Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> exhibited the highest visible-light-driven activity, which could degrade almost all MO within 120 min. Under NIR light irradiation, the Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> composite showed the best photocatalytic activity and decomposed more than 30% of MO after 240 min of irradiation. Recycling experiments confirmed that the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites had superior cycle performance and stability. The photocatalytic activity was highly dependent on the Ag<sub>2</sub>S content and the well-defined core/shell structure. The enhancement in photocatalytic activity of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites could be mainly ascribed to the efficient separation of photogenerated charge carriers between the intimate interface of Ag<sub>2</sub>S shell and Ag<sub>3</sub>PO<sub>4</sub> core. The in-situ formed Ag nanoparticles at surfaces of Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S during photocatalytic process acted as charge transmission bridges and electron trapping centers, respectively, resulting in the stable Ag<sub>2</sub>S/Ag/Ag<sub>3</sub>PO<sub>4</sub> Z-scheme system and Ag/Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> ternary system. Based on the experimental results, the possible photocatalytic mechanisms for MO degradation over Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites under both visible and NIR light irradiation were proposed.

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#### 1. Introduction

Since Fujishima and Honda discovered photocatalytic waterspitting on  $TiO_2$  electrodes for hydrogen production in 1972 [1], research interest in semiconductor photocatalysis has grown significantly owing to its potential applications in environmental purification and solar energy conversion. Yet, the traditional photocatalytic materials, such as  $TiO_2$  and ZnO, require UV light to be activated because of their large bandgap of  $\sim$ 3.2 eV. It is generally known that UV light in the overall solar spectrum takes up only  $\sim$ 5%, whereas the percentages of visible light and near-infrared (NIR) light are about  $\sim$ 49% and  $\sim$ 46%, respectively [2]. For better

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use of solar energy, much effort has been devoted to extending the absorption of UV-active photocatalyst into visible light or NIR light region by adjusting the bandgap through noble metal deposition, metal/nonmetal dopant, coupling with upconversion material and so on [3–6]. On the other hand, some researchers have also paid considerable attention on the exploration of novel photocatalytic materials that are sensitive to visible light or NIR light, such as  $In_{1-x}Ni_xTaO_4$  [7],  $ZnIn_2S_4$  [8],  $g-C_3N_4$  [9], Ag/AgX (X = Cl, Br, I) [10],  $Cu_2(OH)PO_4$  [11], and so on.

Among the reported novel photocatalysts, silver orthophosphate  $(Ag_3PO_4)$  with a suitable band gap  $(2.36\,\mathrm{eV})$  has received more attention in the fields of  $O_2$  evolution from water and organic pollutant treatment under visible light irradiation [12]. However, some limitations inherent in the  $Ag_3PO_4$  photocatalytic system should be paid more attention, such as the high solubility, the photo-corrosion issue and the fast recombination of photogenerated charge carriers (electrons and holes). Many strategies against its drawbacks have been reported. For example, Bi, Ye et al. inves-

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tigated the facet effect on the photocatalytic properties of Ag<sub>3</sub>PO<sub>4</sub> and synthesized single-crystalline Ag<sub>3</sub>PO<sub>4</sub> rhombic dodecahedrons with only active {110} facets exposed [13]. They further fabricated Ag/Ag<sub>3</sub>PO<sub>4</sub> heterostructures to improve the photocatalytic activity of Ag<sub>3</sub>PO<sub>4</sub> [14,15]. Constructing heterostructure/composite photocatalysts has proven to be effective in reducing the recombination of charge carriers and strengthening the structural stability of singe component [16-19]. Many Ag<sub>3</sub>PO<sub>4</sub>-based composites including Ag<sub>3</sub>PO<sub>4</sub>/TiO<sub>2</sub> [20], Ag<sub>3</sub>PO<sub>4</sub>/In(OH)<sub>3</sub> [21], Ag<sub>3</sub>PO<sub>4</sub>/BiVO<sub>4</sub> [22], Ag<sub>3</sub>PO<sub>4</sub>/BiOI [23], Ag<sub>3</sub>PO<sub>4</sub>/SnO<sub>2</sub> [24], Ag<sub>3</sub>PO<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> [25], Ag<sub>3</sub>PO<sub>4</sub>/graphene oxide [26], Ag<sub>3</sub>PO<sub>4</sub>/SiO<sub>2</sub> [27], and so on have been fabricated and shown to be highly efficient in photocatalysis. More specially, the well interfacial connection in composite photocatalysts is considered to be crucial in promoting the photogenerated charge carries transfer and/or separation and thus the photocatalytic performance [28]. A solution to achieve strong interfacial contact is the preparation of composite photocatalysts with a core/shell structure [29]. For instance, Yu et al. fabricated core/shell structured PAN/Ag<sub>3</sub>PO<sub>4</sub> nanofibers by combining electrospinning technology with ion exchange reaction and found that the introduction of PAN significantly enhanced visible light photocatalytic activity [30]. Highly efficient AgX/Ag<sub>3</sub>PO<sub>4</sub> (X = Cl, Br, I) core-shell heterostructures have further been obtained by Bi et al. through the epitaxial growth method [31]. The coating of AgX shell promoted the effective separation of charge carries and simultaneously prevented the dissolution of the Ag<sub>3</sub>PO<sub>4</sub> core-crystals, accounting for excellent catalytic activity and structural stability. The Ag<sub>3</sub>PO<sub>4</sub>@g-C<sub>3</sub>N<sub>4</sub> core-shell composites synthesized by ultrasonication-chemisorption method were also shown to be efficient and stable in organic dye degradation under visible light irradiation [32]. Therefore, the design of core/shell structures between Ag<sub>3</sub>PO<sub>4</sub> and an appropriate photocatalyst is an effective approach to improve the photocatalytic performance (both activity and stability) of Ag<sub>3</sub>PO<sub>4</sub>.

Ag<sub>2</sub>S with a narrow band gap of around 1 eV and a large absorption coefficient in the entire solar spectrum, has received great attention as promising photocatalysts or photosensitizer for photoor photoelectrocatalysts under visible and NIR light irradiation [33-35]. For example, Jiang et al. explored the wide application of Ag<sub>2</sub>S as photocatalyst worked under visible and NIR light irradiation [36]. Li et al. developed the Ag<sub>2</sub>S-Ag-TiO<sub>2</sub> hybrid structures and found that the ternary hybrid exhibited dramatically enhanced performance in photocatalytic hydrogen production under full-spectrum irradiation [37]. More recently, Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites with outstanding visible-light-driven photocatalytic performance have also been fabricated in view of the suitable band structure between Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub>. Tang et al. synthesized Ag@(Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub>) composites via an anion-exchange method and a subsequent calcination treatment [38]. Nevertheless, the hightemperature treatment led to particle agglomeration/coarsening and moreover adverse effects on charge transfer at the ternary interface. To improve the interfacial effect, Ma et al. developed an in situ synthesis strategy to achieve Ag<sub>3</sub>PO<sub>4</sub>/Ag<sub>2</sub>S composites by assembling of Ag<sub>3</sub>PO<sub>4</sub> nanoparticles on two-dimensional Ag<sub>2</sub>S sheets [39]. However, the exposed Ag<sub>3</sub>PO<sub>4</sub> nanoparticles in Ag<sub>3</sub>PO<sub>4</sub>/Ag<sub>2</sub>S composites still suffered from the structural instability issue in aqueous solution. In consideration of the above discussion, constructing of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> core-shell composites may be a feasible strategy to resolve all issues. The cover of Ag<sub>2</sub>S nanoparticles shell on Ag<sub>3</sub>PO<sub>4</sub> core can not only promote the effective separation of charge carries but also improve the structural stability of the composites. Besides, due to the light sensitization effect of Ag<sub>2</sub>S shell, the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> may possess photocatalytic performance under NIR light illumination. To our knowledge, there are no reports concerning the NIR-light-driven photocatalytic performance of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites. Furthermore, the structure–property relationship and the corresponding photocatalytic mechanisms still remain the challenge.

Herein, we firstly report the synthesis of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites with well-defined core/shell structure according to the anion-exchange principle. The highly dispersed Ag<sub>2</sub>S nanoparticles can *in-situ* grow onto the surfaces of Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons. During the degradation of methyl orange (MO) under visible light and NIR light irradiation, the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites exhibit much higher photocatalytic activities than pure Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub>. The activity of the composites is found to be highly dependent on the Ag<sub>2</sub>S content and the core/shell structure. Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites also display excellent stability irradiated under visible and NIR light. The possible photocatalytic mechanisms under both conditions are thereby proposed.

#### 2. Experimental section

#### 2.1. Materials

All reagents were of analytical grade and used without further purification. Silver nitrate (AgNO<sub>3</sub>), hexamethylenetetramine (HMT), sodium dihydrogen phosphate (NaH<sub>2</sub>PO<sub>4</sub>), sodium sulphide (Na<sub>2</sub>S), terepthalic acid (TA), ammonium oxalate (AO), benzoquinone (BQ) and tertbutyl alcohol (TBA) were purchased from Sigma–Aldrich India. Double distilled water was used throughout the study.

#### 2.2. Sample preparation

## 2.2.1. Synthesis of Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons

 $Ag_3PO_4$  dodecahedrons were prepared by a simple precipitation method. In a typical synthesis, 1.87 mmol  $AgNO_3$  was dissolved in 40 mL distilled water. Then 0.315 g HMT was added to the above suspension under intensive stirring and the entire solution became a milky white color. After stirring 15 min,  $NaH_2PO_4$  aqueous solution (0.15 M) was added with drop by drop to the above solution, and golden yellow precipitation would be formed. The as-obtained precipitate was washed and kept in an oven at 60 °C.

#### 2.2.2. Synthesis of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites

 $\rm Ag_2S/Ag_3PO_4$  composites were obtained through an  $\it in\text{-}situ$  anion exchange method in dark conditions at room temperature. The obtained  $\rm Ag_3PO_4$  (1.0 g) powder was dispersed in 50 mL of distilled water and suspension was ultrasonicated for 0.5 h. Subsequently certain amounts of Na\_2S solutions (1–80 mL, 0.01 M) were added drop wise into the above Ag\_3PO\_4 suspension with constant stirring. The obtained suspension was vigorously stirred for 4 h. The precipitates were collected, washed with deionized water several times and finally dried at 60 °C for 24 h. The series of Ag\_2S/Ag\_3PO\_4 composites were obtained by varying the concentration of Na\_2S aqueous solution and denoted Ag\_2S-X%, where X% is the molar ratio of S relative to P. The pure Ag\_2S samples were obtained by a simple precipitation reaction between Na\_2S solutions and AgNO\_3 solutions at room temperature.

#### 2.3. Sample characterizations

X-ray diffraction patterns (XRD) were collected on a Rigaku MinFlex II equipped with Cu  $K\alpha$  irradiation. BET surface area was performed on a Micromeritics ASAP 2000 analyzer. Morphologies and composition of the samples were observed by field emission scanning electron microscope (FE-SEM) (Hitachi S-4800), transmission electron microscope (TEM) (JEM-2010) and energy dispersive X-ray spectroscopy (EDX). The content of S and P was quantified on atomic fluorescence spectroscopy (AFS, AFS-9600). X-ray photoelectron spectra (XPS) were recorded on an ESCALAB

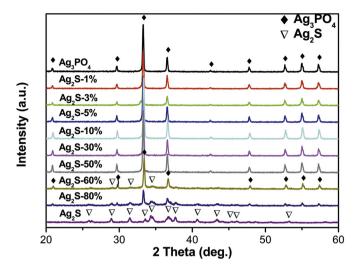


Fig. 1. XRD patterns of pure Ag<sub>3</sub>PO<sub>4</sub>, Ag<sub>2</sub>S and Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites.

250 photoelectron spectroscope (Thermo Fisher Scientific) with monochromatic Al  $K\alpha$  radiation (E=1486.2 eV). UV-vis-NIR diffuse reflectance spectra (DRS) were collected using a Varian Cary 500 Scan system equipped with an integrating sphere attachment. Steady-state photoluminescence (PL) spectra of the samples were performed on an Edinburgh Analytical Instruments F-4600 coupled with a time correlated single-photo counting system. Transient PL spectra were performed on an Edinburgh Analytical Instruments FLSP-920 using a 375 wavelength pulse laser (pulse width 70 ps). Electrochemical impedance (EIS) was recorded on a CHI-660B electrochemical workstation (Chenhua Co.) in a standard three-electrode configuration.

#### 2.4. Evaluation of photocatalytic activity

Photocatalytic experiments were performed in an aqueous solution at ambient temperature irradiated by visible light (300 W Xe arc lamp) and NIR light (300 W infrared lamp), which was respectively obtained by using cutoff filters to remove light of  $\lambda$  < 400 nm or  $\lambda$  < 760 nm. Briefly, 35 mg of photocatalyst was suspended in 80 mL aqueous solution of MO dye (10 ppm). Prior to irradiation, the suspension was magnetically stirred in dark for 0.5 h to establish an adsorption-desorption equilibrium. A 4 mL aliquot was taken at several minutes intervals during the experiment and centrifuged to remove the powders. The residual concentration of dye was analyzed on a Shanghai Youke UV756CRT spectrophotometer. The degradation percentage is reported as  $C/C_0$ , where  $C_0$  is the initial concentration of dye, and C represents the corresponding concentration at a certain time interval. The stability was tested as follows: after each MO degradation reaction, the suspension was filtered and the solids were washed with water and dried at 60 °C in air. Then the regenerated product was employed to degrade a new 10 ppm MO aqueous solution for another test under the same visible light irradiation. During the radical-trapping experiments, 35 mg catalyst with TBA (1 mL) or AO (11.7 mg) or BQ (1.2 mg) was dispersed in MO aqueous solution (80 mL, 10 ppm) before visible light irradiation.

#### 3. Results and discussion

### 3.1. Texture properties

The phase composition and crystal properties of the obtained samples are analyzed by XRD. Fig. 1 shows the XRD patterns of the  $Ag_3PO_4$ ,  $Ag_2S$  and  $Ag_2S/Ag_3PO_4$  composites. It can be observed that the diffraction peaks of  $Ag_3PO_4$  are indexed to the cubic phase of

**Table 1**The specific surface areas of the as-prepared samples.

Samples	Specific surface area (m <sup>2</sup> /g)	The molar ratio of S/P (%)
Ag <sub>3</sub> PO <sub>4</sub>	0.8	0
Ag <sub>2</sub> S-3%	1.1	2.56
Ag <sub>2</sub> S-5%	2.4	4.02
Ag <sub>2</sub> S-30%	4.5	23.74
Ag <sub>2</sub> S-50%	4.8	38.24
Ag <sub>2</sub> S-80%	4.7	59.36
$Ag_2S$	7.2	_

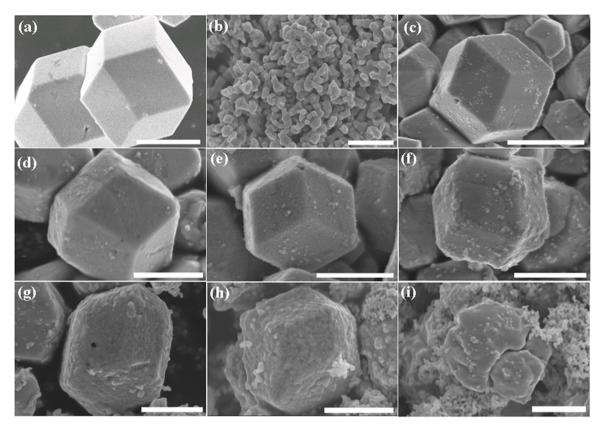
Ag<sub>3</sub>PO<sub>4</sub> crystal (ICPDS No. 06-0505) and no other diffraction peaks are detected, which indicates that the produced Ag<sub>3</sub>PO<sub>4</sub> has high purity without other impurities. Compared with Ag<sub>3</sub>PO<sub>4</sub>, Ag<sub>2</sub>S displays relatively weak and broad diffraction peaks. These peaks are attributed to the monoclinic structure (ICPDS No.14-0072). The broad diffraction peaks of Ag<sub>2</sub>S are attributed to the relatively small crystalline size [36,40]. As for the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites, no obvious diffraction peaks of Ag<sub>2</sub>S are observed in Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> when the amount of Ag<sub>2</sub>S is lower than 50%. This may be due to the fact that Ag<sub>2</sub>S nanoparticles are highly dispersed on the surface of Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons. However, with the increase of the loading amount of Ag<sub>2</sub>S above 60%, the intensity of diffraction peaks of Ag<sub>2</sub>S increases whereas that of Ag<sub>3</sub>PO<sub>4</sub> decreases simultaneously. It is notable that no characteristic peaks attributed to metal silver and other new compounds are detected in all Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> samples, indicating that Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub> maintain as pure phase and no impurities formed in Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites.

The BET surface area of the pure samples ( $Ag_3PO_4$  and  $Ag_2S$ ) and  $Ag_2S/Ag_3PO_4$  composites has been measured. From Table 1, we can see that pure  $Ag_3PO_4$  exhibits small specific surface area of  $0.8~m^2/g$  while pure  $Ag_2S$  have a slightly large value of  $7.2~m^2/g$ . The growth of  $Ag_2S$  onto  $Ag_3PO_4$  has little influence on the BET surface area of the composites, which indicates the negligible effect of surface area on the followed photocatalytic activity. The actual molar ratio of S/P in the  $Ag_2S/Ag_3PO_4$  composites is determined by AFS and the results are shown in Table 1.

# 3.2. Microstructures

The SEM images of Ag<sub>3</sub>PO<sub>4</sub>, Ag<sub>2</sub>S as well as Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites are presented in Fig. 2. Fig. 2a shows that Ag<sub>3</sub>PO<sub>4</sub> samples appear with the homogeneous dodecahedral morphology with an average diameter of 2 µm. It can be clearly observed that the Ag<sub>3</sub>PO<sub>4</sub> particles have clean and smooth surfaces. As compared, pure Ag<sub>2</sub>S particles are irregular aggregates which are composed of crystal grains with diameters of 30-80 nm (Fig. 2b, Fig. S1), consistent with the reported result [36]. In the case of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites, novel core/shell heterocrystals consisting of Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons core and Ag<sub>2</sub>S nanoparticles shell are obtained by the anion-exchange process (Fig. S2), similar to the AgX (X = Cl, Br, I)/Ag<sub>3</sub>PO<sub>4</sub> heterocrystals reported by Bi et al. [31]. It should be noted that the core/shell structure and the size of Ag<sub>2</sub>S nanoparticles are highly dependent on the amount of  $S^{2-}$ . When a low amount (1-5%) of Na<sub>2</sub>S is introduced, tiny and uniform Ag<sub>2</sub>S nanoparticles are highly dispersed on the surface of Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons (Fig. 2c-e). With increasing the Na<sub>2</sub>S amount (10-50%), more and large Ag<sub>2</sub>S nanoparticles are covered on the Ag<sub>3</sub>PO<sub>4</sub> surfaces (Fig. 2f-h). Further increasing the Na<sub>2</sub>S amount (60–80%) obviously destroys the well-defined core/shell structure. Instead, only broken Ag<sub>3</sub>PO<sub>4</sub> particles and detached Ag<sub>2</sub>S aggregates can be found in these composites (Fig. 2i and Fig. S3).

The microstructure and composition of the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites are further investigated by TEM and EDX. The TEM images for the Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> sample show that a few of Ag<sub>2</sub>S nanopar-



**Fig. 2.** SEM images of (a) pure  $Ag_3PO_4$ , (b) pure  $Ag_2S_4$ , and  $Ag_2S_4Ag_3PO_4$  composites: (c)  $Ag_2S_5$ , (d)  $Ag_2S_5$ , (e)  $Ag_2S_5$ , (f)  $Ag_2S_5$ , (g)  $Ag_2S_5$ , (i)  $Ag_2S_5$ , (i)  $Ag_2S_5$ , (all scale bars are 1  $\mu$ m).

ticles with size below 5 nm are deposited on the surface of Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons (Fig. 3a and b and Fig. S4a and b). Due to the limited amount of S<sup>2-</sup>, the Ag<sub>3</sub>PO<sub>4</sub> surface is partially covered by Ag<sub>2</sub>S nanoparticles, shown as the diagram insert in Fig. 3b. The EDX spectrum (Fig. 3c) indicates that the Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> sample contains Ag, P, S and O elements, confirming the presence of Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub>. For Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> sample, it can be observed that Ag<sub>2</sub>S nanoparticles with increasing amounts and large sizes ranging from approximately 20 to 50 nm are uniformly distributed on the Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons (Fig. 3d and e and Fig. S4c and d), resulting in the fully covered core/shell structures (shown as the diagram in Fig. 3e). The EDX spectrum (Fig. 3f) also proves the coexistence of Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S. Accordingly, the well-defined Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> core/shell composites can be fabricated by the anionexchange route. That is, S<sup>2-</sup> ions substitute for the [PO<sub>4</sub>]<sup>3-</sup> units on the surface of Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons to form Ag<sub>2</sub>S nanoparticles, because the much smaller  $K_{sp}$  value  $(6.6 \times 10^{-50})$  of Ag<sub>2</sub>S than  $Ag_3PO_4(1.4 \times 10^{-16})$  easily facilitates this conversion [41]. The nucleation and growth of Ag<sub>2</sub>S nanoparticles is highly dependent on the concentration of  $S^{2-}$  during this ion-exchange process.

XPS analyses are carried out on the typical  $Ag_2S-5\%/Ag_3PO_4$  and  $Ag_2S-5\%/Ag_3PO_4$  composites to evaluate the chemical valence state of the elements, and the results are shown in Fig. 4. The full-scale XPS pattern is shown in Fig. 4a, which indicates the existence of Ag, O, P and S elements in both samples. The high resolution XPS spectra of Ag 3d show that there are two peaks located at 367.8 and 373.8 eV (Fig. 4b), which can be respectively attributed to Ag  $3d_{5/2}$  and Ag  $3d_{3/2}$  binding energies in  $Ag_3PO_4$  or  $Ag_2S$  [13,40,42]. Peak fitting of Ag 3d spectra indicates the coexistence of metallic  $Ag^0$  and  $Ag^+$  in both composites. The relative percentages of metallic  $Ag^0$  in  $Ag_2S-5\%/Ag_3PO_4$  and  $Ag_2S-5\%/Ag_3PO_4$  are estimated to be 7.5% and 23.3%, respectively, suggesting that more metallic  $Ag^0$  are formed in the high concentration of  $S^{2-}$  system. In the O 1s spectra

(Fig. 4c), the main peak at  $530.7\,\mathrm{eV}$  is assigned to the oxygen in the  $Ag_3PO_4$  crystal lattice and the weak peak at  $532.7\,\mathrm{eV}$  is due to the chemisorbed  $H_2O$  or  $OH^-$  on the surface of the samples. The P 2p peak is located at  $132.5\,\mathrm{eV}$  (Fig. 4d), affirming that the valence state of P is +5. In the case of the S 2p spectra, the two S 2p<sub>3/2</sub> and S 2p<sub>1/2</sub> peaks located at around 161.8 and 163.0 eV are characteristics of sulfur anions in the lattice of  $Ag_2S$  correspondingly [36,43,44]. It is apparent noted that the peak intensity of P 2p in  $Ag_2S$ -5%/ $Ag_3PO_4$  is much stronger than that in  $Ag_2S$ -5%/ $Ag_3PO_4$ , while that of S 2p in  $Ag_2S$ -5%/ $Ag_3PO_4$  is weaker than that in  $Ag_2S$ -50%/ $Ag_3PO_4$ . This result evidently suggests that more  $Ag_2S$  nanoparticles are deposited on the  $Ag_3PO_4$  surface with increasing  $S^{2-}$  amount, quite consistent with the SEM and TEM results.

# 3.3. Optical properties

The optical properties of the samples are investigated and the corresponding DRS spectra are shown in Fig. 5a. Ag<sub>3</sub>PO<sub>4</sub> dodecahedrons exhibit strong absorption in the UV and visible light region. A steep absorption edge is located at approximately 520 nm. Ag<sub>2</sub>S nanoparticles exhibit a remarkable absorption in the entire UV-vis-NIR region. The absorption edge can be extended to around 1400 nm, indicating that it can significantly adsorb more light in the visible-NIR region as compared with pure Ag<sub>3</sub>PO<sub>4</sub>. Accordingly, for the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites, one can see that the absorption edge demonstrates a gradual red-shift from 520 nm to 1500 nm with the increase of Ag<sub>2</sub>S contents. The obvious red-shift of absorption edge for Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> samples is attributed to the excellent sensitization effect of Ag<sub>2</sub>S, which can absorb more visible and NIR light and make the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites effectively harness the full spectrum of solar light to strengthen the photocatalytic activities. Moreover, the majority of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites are found to show much larger light absorption than Ag<sub>2</sub>S. This might be

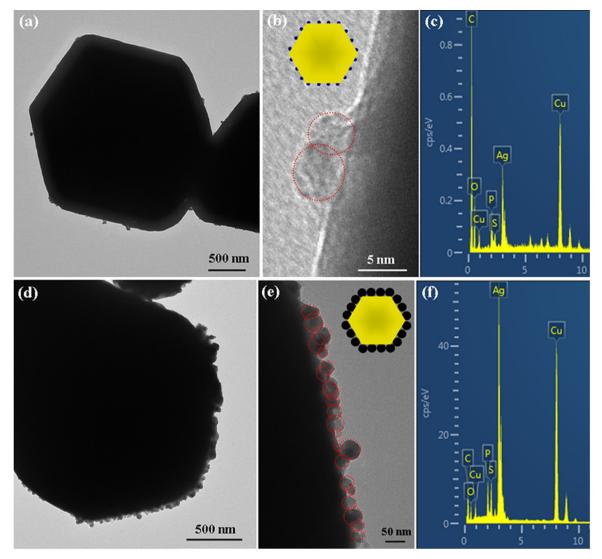


Fig. 3. TEM and EDX pattern of (a-c) Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> and (d-f) Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> composites.

attributed to the plasmon resonance of metallic Ag<sup>0</sup> *in-situ* formed during ion-exchange process [10,45].

The band gap energy  $(E_g)$  of  $Ag_2S$  and  $Ag_3PO_4$  is calculated by the following equation:  $\alpha h \nu = A(h\nu - E_g)^{n/2}$ , where  $\alpha$ , h,  $\nu$  and A are the absorption coefficient, Planck constant, light frequency and proportionality, respectively; n is decided from the properties of the transition in a semiconductor. Ag<sub>3</sub>PO<sub>4</sub> is an indirect band semiconductor, and the value of n is 4;  $Ag_2S$  is reported to be a direct band semiconductor [46], and the value of *n* is 1. As shown in Fig. 5b and c, the band gaps of Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S are about 2.33 and 0.99 eV, respectively, which are in accordance with the previous reports [12,42]. Moreover, it is known that the band edge positions of the conduction band (CB) and the valence band (VB) of photocatalysts also play a crucial role in determining their photocatalytic activity and can be achieved by the empirical formula:  $E_{VB} = X - E_0 + 1/2E_g$ and  $E_{CB} = X - E_0 - 1/2E_g$ , in which X is the absolute electronegativity of the semiconductor, determined by the geometric mean of the absolute electronegativity of constituent atoms, which is defined as the arithmetic mean of the atomic electron affinity and the first ionization energy;  $E_0$  is the energy of free electrons on the hydrogen scale (about 4.5 eV); and  $E_g$  is the band gap of the semiconductor. Accordingly, the  $E_{CB}$  and  $E_{VB}$  of Ag<sub>3</sub>PO<sub>4</sub> are determined to be 0.25 and 2.67 eV, while those of  $Ag_2S$  are -0.025 and 0.965 eV. The appropriate energy level of the CB and VB edges assures that Ag<sub>2</sub>S can act as sensitizers, which makes Ag<sub>3</sub>PO<sub>4</sub> highly active under visible light and NIR light.

# 3.4. Photocatalytic activity

As indicated above, the results from full solar spectrum absorption of the  $\rm Ag_2S/Ag_3PO_4$  composites predict their enhanced photocatalytic activity in wide spectral range of solar light. Herein, the photocatalytic activity of the  $\rm Ag_2S/Ag_3PO_4$  composites is investigated for the decomposition of MO under visible and NIR light irradiation, respectively.

# 3.4.1. Visible-light-driven photocatalytic property

As shown from Fig. 6a, pure Ag<sub>3</sub>PO<sub>4</sub> and all Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites display negligible adsorption capability for MO due to the low surface area and negatively charged surface of Ag<sub>3</sub>PO<sub>4</sub> [27]. Ag<sub>2</sub>S nanoparticles show obvious adsorption capability (less than 20%) for MO because of the relatively high surface area. However, no decomposition of MO is observed in the presence of Ag<sub>2</sub>S under visible light irradiation while pure Ag<sub>3</sub>PO<sub>4</sub> can degrade 60% of MO after irradiation of 120 min. As compared, the Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> sample exhibits much higher activity with respect to individual Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub>; the degradation of MO reaches nearly 100% after 120 min light irradiation. In addition, the physical mixture

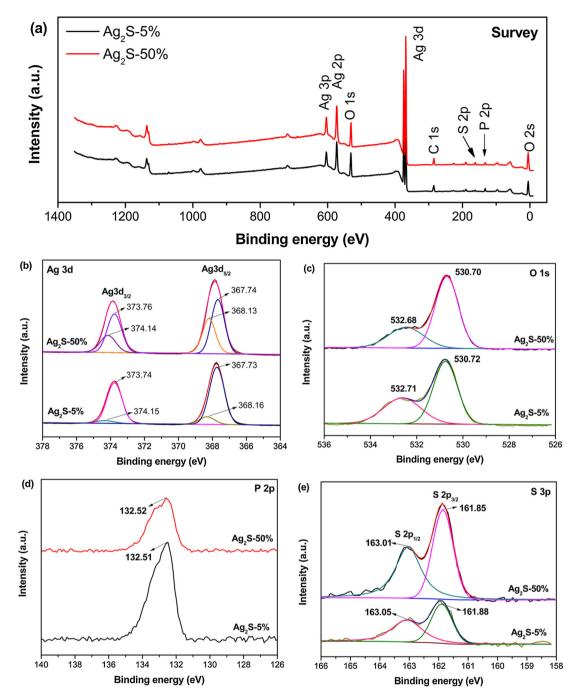


Fig. 4. XPS spectra of (a) survey (b) Ag 3d, (c) O 1s, (d) P 2p and (e) S 2p of the Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> composites.

of  $Ag_2S$  and  $Ag_3PO_4$  at same  $Ag_2S$  content (5%) shows a lower photocatalytic activity compared to the ion-exchange synthesized  $Ag_2S-5\%/Ag_3PO_4$  composite. This may be attributed to the fact that the physically mixing of  $Ag_2S$  and  $Ag_3PO_4$  results in poor contact between these two components, which hinders the interface charge transfer [47]. The photolysis of MO in the absence of photocatalyst is negligible within the test period as MO molecule is quite stable under visible light irradiation. No decomposition of MO is observed in dark even with catalyst, implying the importance of light irradiation.

The photocatalytic degradation kinetics towards the degradation of MO over various samples was investigated by fitting the experimental data to the Langmuir–Hinshelwood model. Since the concentration is low, the pseudo-first-order kinetic model might

be assumed as  $-\ln(C/C_0) = kt$ , where  $C_0$  and C are the balanced and the final dye concentration, respectively, k is the pseudo-first order rate constant and t is light irradiation time. The results indicate that the reaction kinetics for all of the catalysts could be fitted to the pseudo-first-order kinetic model as presented in Fig. 6b. The visible-light-driven activity of  $Ag_2S/Ag_3PO_4$  is highly dependent on the  $Ag_2S$  content. The  $Ag_2S/Ag_3PO_4$  composites with low amount (1-5%) of  $Ag_2S$  show enhanced photocatalytic activity; however, excess  $Ag_2S$  (10-80%) result in the decrease of the photocatalytic activity (Fig. 6c). Among all composites,  $Ag_2S-5\%/Ag_3PO_4$  exhibits the most prominent photocatalytic activity and the maximum rate constant  $(0.03388 \, \text{min}^{-1})$ , which is  $\sim 4 \, \text{and} \, 176 \, \text{times}$  as high as that of  $Ag_3PO_4$   $(0.00839 \, \text{min}^{-1})$  and  $Ag_2S$  (1.91883E-4), respectively. The  $Ag_2S-5\%/Ag_3PO_4$  composite also displays superior photocat-

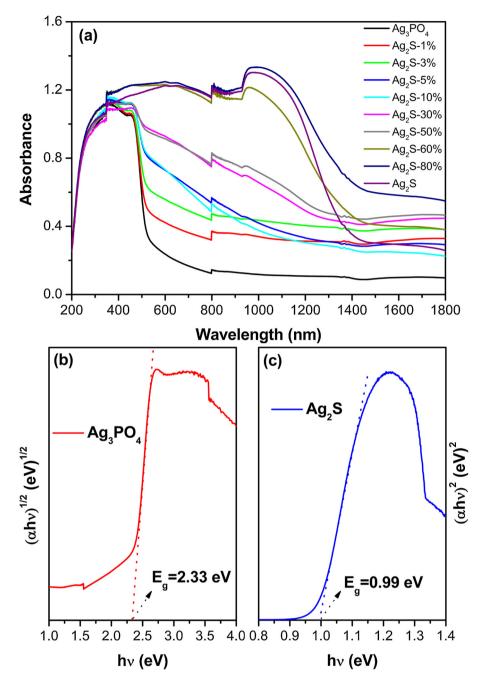


Fig. 5. (a) DRS spectra of Ag<sub>3</sub>PO<sub>4</sub>, Ag<sub>2</sub>S, and Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites, (b and c) band gap energies of Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S.

alytic performance to  $Ag_3PO_4$  towards the degradation of colorless chemical pollutants (p-chlorophenol and salicylic acid) under visible light irradiation (Fig. S5). As a result of quenching experiments shown in Fig. 6d, the photocatalytic degradation efficiency is largely suppressed after adding AO and BQ, indicating that  $h^+$  and  $O_2^{\bullet-}$  are the main active species over  $Ag_2S-5\%/Ag_3PO_4$  in the visible-light-driven photocatalytic process.

In addition to photocatalytic activity, stability is regarded as another important issue determining the practical application of photocatalysts, especially for Ag-based photocatalysts. Therefore, the cyclic degradation of MO under visible light over pure Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> composite was carried out. It can be seen that pure Ag<sub>3</sub>PO<sub>4</sub> displayed an obvious deactivation during the repeated three experiments (Fig. S6). As compared, the Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> composite exhibited excellent photocatalytic stability. As shown in Fig. 7a, the activity of Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> reaches 99% after one

cycle (120 min) and remains at above 97% throughout the second and third cycles, indicating that Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> composite acts as a stable photocatalyst. To check the phase of the used composite, XRD analysis is carried out on Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> after three times recycle experiments. The XRD pattern in Fig. 7b shows that a new diffraction peak assigning to Ag appears at 38.06°, indicating the formation of metallic Ag during photocatalytic reaction. The microstructure and composition of Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> after reaction are further investigated by TEM and EDX. From Fig. 7c, it can be clearly observed that many nanoparticles with size of about 40 nm are loaded on the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> surface. These outside nanoparticles are only composed of Ag element (red region in Fig. 7c), while the inner samples consist of Ag, S, P and O elements (blue region in Fig. 7c), as evident by EDX in Fig. 7d and e. The Ag 3d XPS of used Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> displays characteristic peaks attributing to metallic Ag and Ag<sup>+</sup> (Fig. S7). The calculated content of metallic Ag in the

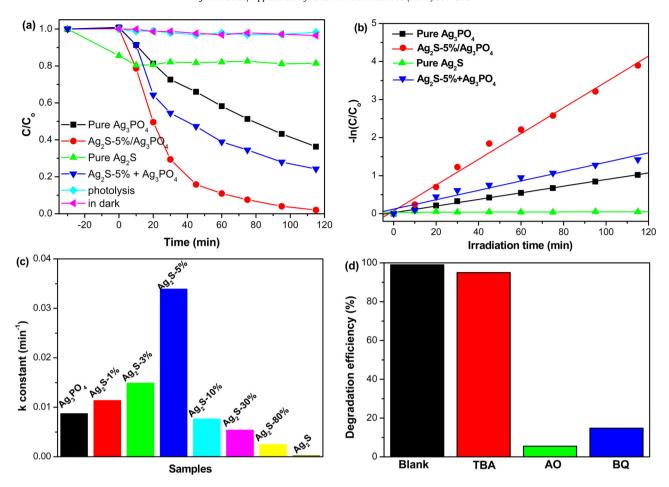
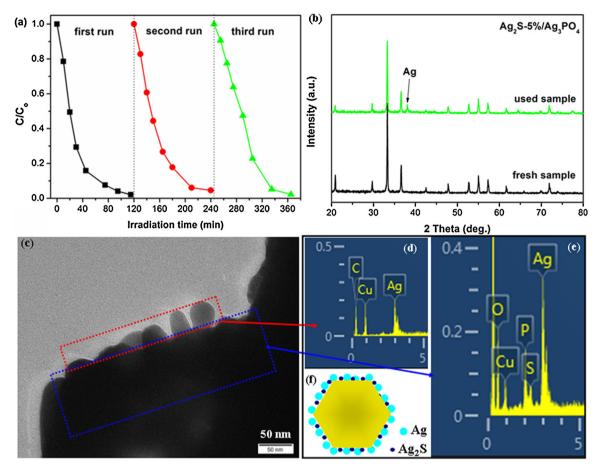


Fig. 6. (a) Photocatalytic degradation and (b) kinetic fit curves for the degradation of MO over various samples under visible light irradiation, (c) the corresponding rate constants, (d) effect of scavengers on the degradation efficiency of MO over  $Ag_2S-5\%/Ag_3PO_4$  under visible light irradiation.

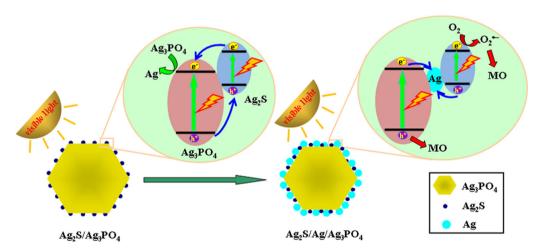
used sample is about 13.3%, much higher than that in the fresh sample (7.5%). All these results show that some  $Ag^+$  in  $Ag_2S/Ag_3PO_4$  is reduced to  $Ag^0$  during the photocatalytic experiments and a typical ternary photocatalyst  $Ag_2S/Ag/Ag_3PO_4$  is in-situ formed (shown as the diagram in Fig. 7f).

On the basis of the above-mentioned results, a possible structural change and mechanism for MO degradation of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> under visible light irradiation is proposed and shown in Scheme 1. Under visible light irradiation, both Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub> are simultaneously excited and generate electron-hole pairs. The photoexcited electrons on more negative CB of Ag<sub>2</sub>S (-0.025 eV) prefer to flow down to the CB of Ag<sub>3</sub>PO<sub>4</sub> (0.25 eV) while the photoexcited holes would transfer from more positive VB of Ag<sub>2</sub>S (0.965 eV) to the VB of Ag<sub>3</sub>PO<sub>4</sub> (2.67 eV), resulting in an efficient charge separation process in Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub>. The electrons accumulated on the CB of Ag<sub>3</sub>PO<sub>4</sub> are prone to combine with mobile interstitial silver ions from Ag<sub>3</sub>PO<sub>4</sub> surfaces in the early photocatalytic process, leading to the formation of metallic Ag species and transforming the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> into Ag<sub>2</sub>S/Ag/Ag<sub>3</sub>PO<sub>4</sub> ternary system eventually, like the reported Ag<sub>3</sub>PO<sub>4</sub>/Ag/C<sub>3</sub>N<sub>4</sub>, Ag<sub>3</sub>PO<sub>4</sub>/Ag/TiO<sub>2</sub>, Ag<sub>3</sub>PO<sub>4</sub>/Ag/Ag/ [25,37,48–50]. In these ternary systems, the in-situ formed Ag nanoparticles usually act as the charge transmission bridges, which can efficiently promote the separation of electron-hole pairs by a Z-scheme mechanism and ultimately enhance the photocatalytic performance. Our recent study on Ag<sub>3</sub>PO<sub>4</sub>/In<sub>2</sub>S<sub>3</sub> also proved the formation of Ag<sub>3</sub>PO<sub>4</sub>/Ag/In<sub>2</sub>S<sub>3</sub> Z-scheme system [39]. On the basis of the radicaltrapping experiments and the similarity between Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>3</sub>PO<sub>4</sub>/In<sub>2</sub>S<sub>3</sub>, we believe that the Z-scheme mechanism might also work in the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites under visible light irradiation.

Driven by the transmission bridge role of Ag nanoparticles, the electrons in CB of Ag<sub>3</sub>PO<sub>4</sub> and the holes in VB of Ag<sub>2</sub>S shift to metallic Ag simultaneously, which enhances the separation of electron-hole pairs and enables the holes and electrons to remain on Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S, respectively, accounting for the structural stability of the composites. The accumulated electrons in CB of Ag<sub>2</sub>S can further reduce O<sub>2</sub> to initiate the yield of O<sub>2</sub>•- reactive species which can decompose the MO pollutant. At the same time, the holes left on VB of Ag<sub>3</sub>PO<sub>4</sub> with strong oxidation ability may oxidize OH<sup>-</sup> or H<sub>2</sub>O to yield •OH. However, the present experimental results show that no •OH species were generated in Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> system by the photoluminescence (PL) technology with terephthalic acid (TA) as a probe molecule (Fig. S8). Thiven by the transmission bridge role of Ag nanoparticles, the electrons in CB of Ag<sub>3</sub>PO<sub>4</sub> and the holes in VB of Ag<sub>2</sub>S shift to metallic Ag simultaneously, which enhances the separation of electron-hole pairs and enables the holes and electrons to remain on Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S, respectively, accounting for the structural stability of the composites. The accumulated electrons in CB of Ag<sub>2</sub>S can further reduce O<sub>2</sub> to initiate the yield of O<sub>2</sub>•- reactive species which can decompose the MO pollutant. At the same time, the holes left on VB of Ag<sub>3</sub>PO<sub>4</sub> with strong oxidation ability may oxidize OH<sup>−</sup> or H<sub>2</sub>O to yield •OH. However, the present experimental results show that no •OH species were generated in Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> system by the photoluminescence (PL) technology with terephthalic acid (TA) as a probe molecule (Fig. S8). This declares that the holes reactive species would directly participate in the oxidation of MO.



**Fig. 7.** (a) The cycling runs in photocatalytic degradation of MO over  $Ag_2S-5\%/Ag_3PO_4$  under visible light irradiation; (b) XRD patterns, (c) TEM and (d and e) EDX patterns of  $Ag_2S-5\%/Ag_3PO_4$  after photocatalytic reaction, (f) the diagram for the  $Ag_2S/Ag/Ag_3PO_4$  system. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)



 $\textbf{Scheme 1.} \ \ \text{The proposed structural change and mechanism for MO degradation of } Ag_2S/Ag_3PO_4 \ \ \text{under visible light irradiation}.$ 

# 3.4.2. NIR-light-driven photocatalytic property

In this work, except for the visible light photocatalytic property, we find Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites also possess excellent NIR-light-driven photocatalytic activity for MO. As shown from Fig. 8a, the Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> composite displays superior activity to pure Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub>, as well as the physically mixed Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub>. The photocatalytic reaction for the catalysts is all fitted to the pseudo-first-order kinetic model (Fig. 8b). Similarly, the NIR-light-driven

activity of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> is also closely related to the Ag<sub>2</sub>S contents. As compared to pure Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub>, excellent photocatalytic activities are observed on the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites with Ag<sub>2</sub>S loading of 10–50% (Fig. 8c). Among these composites, the Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> sample exhibits the highest activity and rate constant (0.00143 min<sup>-1</sup>). Fig. 8d displays the trapping experiment of active species for MO degradation over Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> under NIR light

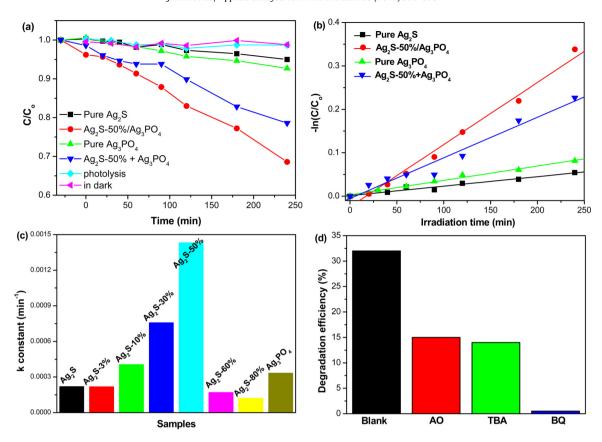


Fig. 8. (a) Photocatalytic degradation and (b) kinetic fit curves for the degradation of MO over various samples under NIR light irradiation, (c) the corresponding rate constants, (d) effect of scavengers on the degradation efficiency of MO over Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> under NIR light irradiation.

irradiation. It can be concluded that the main active species are  $O_2^{\bullet-}$ ,  $h^+$  and  $\bullet$ OH.

The photocatalytic stability of Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> composite is also investigated for the cyclic degradation of MO under NIR light irradiation. As shown in Fig. 9a, after three continuous cycles of photodegradation, the Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> catalyst does not exhibit obvious loss of activity, demonstrating its good recyclability. The XRD patterns (Fig. 9b) reveal that the catalyst collected after the photocatalytic experiments keeps the main Ag<sub>3</sub>PO<sub>4</sub> phase and has an additional small peak assigning to metallic Ag. The used Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> composite inherits the original core/shell structure (Fig. 9c). From the enlarged TEM image in Fig. 9d, we can see that some tiny nanoparticles with average size of 15 nm are dispersed on the surfaces of large Ag<sub>2</sub>S particles. The EDX spectrum (Fig. 9e) indicates that these tiny nanoparticles are mainly composed of Ag element. The increased content (73.1%) of metallic Ag in the used Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub> composite calculated from the Ag 3d XPS spectrum also supports the new formation of metallic Ag nanoparticles (Fig. S9). Considering the full cover of Ag<sub>3</sub>PO<sub>4</sub> by Ag<sub>2</sub>S, it is inferred that the newly formed tiny nanoparticles are metallic Ag, which originate from the in-situ photoreduction of Ag<sub>2</sub>S. Thus, a new Ag/Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> system is formed during the photocatalytic process (Fig. 9f).

A possible structural change and photocatalytic mechanism of  $Ag_2S/Ag_3PO_4$  under NIR light irradiation is also proposed in Scheme 2.  $Ag_2S$  with a narrow band gap energy (0.99 eV) can be easily excited by NIR light ( $\lambda > 760$  nm) and induce the generation of electrons and holes. In the case of  $Ag_3PO_4$ , it cannot be excited by NIR irradiation with energy less than 2.75 eV. The matched band structure between  $Ag_2S$  and  $Ag_3PO_4$  favored the electrons transfer from  $Ag_2S$  to  $Ag_3PO_4$ . However, the full cover of  $Ag_2S$  nanoparticles on  $Ag_3PO_4$  surfaces effectively protects it from dissociation into

interstitial silver ions, inhibiting the reduction of  $Ag_3PO_4$  to metallic Ag by photogenerated electrons. Instead, the gathered electrons on  $Ag_3PO_4$  can react with adsorbed  $O_2$  to yield  $H_2O_2$  and then  $\bullet$ OH [31]. The generation of  $H_2O_2$  over  $Ag_2S$ -50%/ $Ag_3PO_4$  under NIR light irradiation was determined by the DPD method [51]. The results show that the photogeneration of  $H_2O_2$  occurred throughout the process and its concentration increased rapidly within the first 30 min and reached a plateau value after approximately 150 min (Fig. S10). Alternatively, the photogenerated electrons over  $Ag_2S$  could also react with itself to form  $Ag^0$  (Fig. 9d) [42]. The formed  $Ag^0$  species will function as electrons traps to capture photogenerated electrons from CB of  $Ag_2S$  and react with adsorbed  $O_2$  to generate  $O_2$ •-. These generated  $O_2$ •- and  $\bullet$ OH radicals, together with holes left at the VB of  $Ag_2S$ , act as the main active species and finally degrade dye molecules (Fig. 8d).

It is concluded that the improved photocatalytic activity (both visible and NIR light) of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites is highly dependent on the well-defined core/shell structure. The well interfacial contact in composite photocatalysts greatly promotes the charge carries transfer and separation and thus the photocatalytic performance. To justify this assumption, we use PL spectroscopy to survey the extent of charge separation of the pure Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites. Steady-state PL spectra were first investigated and the results are shown in Fig. 10a. Pure Ag<sub>3</sub>PO<sub>4</sub> show a broad emission band ranging from 480 to 600 nm. The strong emission peak at around 550 nm, which has a photon energy approximately equal to the band gap of Ag<sub>3</sub>PO<sub>4</sub>, is considered to originate from the recombination of photogenerated electrons and holes [47,52]. It can be observed that the PL spectra of both Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites are similar to that of pure Ag<sub>3</sub>PO<sub>4</sub>, but the overall emission intensity is significantly weakened, which implies that the recombination of photogenerated electrons and holes is

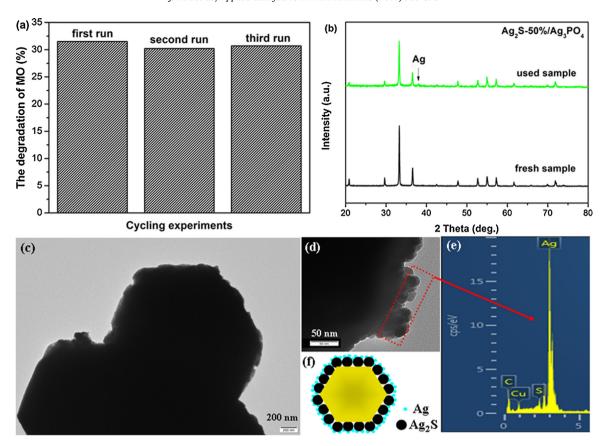
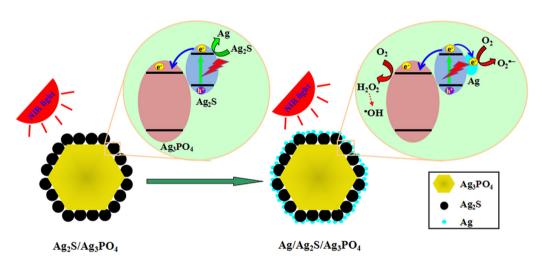


Fig. 9. The cycling runs in photocatalytic degradation of MO over  $Ag_2S-50\%/Ag_3PO_4$  under NIR light irradiation; (b) XRD patterns, (c and d) TEM and (e) EDX patterns of  $Ag_2S-50\%/Ag_3PO_4$  after photocatalytic reaction, (f) the diagram for the  $Ag/Ag_2S/Ag_3PO_4$  system.

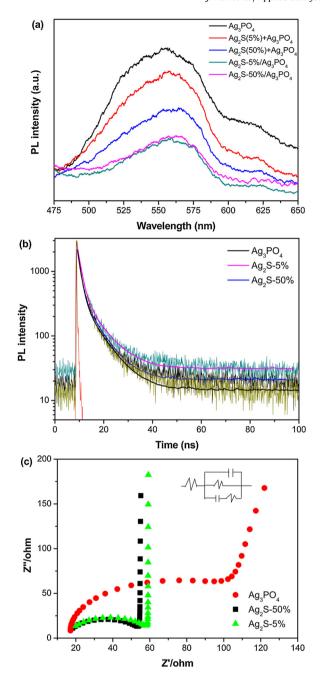


 $\textbf{Scheme 2.} \ \ \text{The proposed structural change and mechanism for MO degradation of } Ag_2S/Ag_2PO_4 \ under \ NIR \ light irradiation.$ 

effectively prevented in the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites [53]. As compared to the physical mixture of Ag<sub>2</sub>S and Ag<sub>3</sub>PO<sub>4</sub>, the Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites also display decreased PL intensity, further suggesting the important feature of core/shell structures in enhancing charge carries separation. Time-resolved PL spectra were further performed. As shown in Fig. 10b, the average life time for pure Ag<sub>3</sub>PO<sub>4</sub> was 3.17 ns while it increased to 3.62 ns and 3.24 ns for Ag<sub>2</sub>S-5%/Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S-50%/Ag<sub>3</sub>PO<sub>4</sub>, respectively. A longer lifetime of the excited state means the photogenerated charge carrier can go further before it decays back to the ground state, leading to more efficient electron-hole separation. Fig. 10c shows the EIS spectra obtained at the interface between the electrolyte and

photocatalysts. The diameters of the arc radii of  $Ag_2S-5\%/Ag_3PO_4$  and  $Ag_2S-5\%/Ag_3PO_4$  are clearly much smaller than that of pure  $Ag_3PO_4$ , also indicating a higher efficiency of charge transfer on the surface of the  $Ag_2S/Ag_3PO_4$  composites.

On the other hand, the  $Ag_2S$  content is also crucial to influence the photocatalytic activity of  $Ag_2S/Ag_3PO_4$  composites. Under visible light irradiation, since both  $Ag_2S$  and  $Ag_3PO_4$  need to be excited, a partial cover of  $Ag_2S$  over  $Ag_3PO_4$  surface is necessary and thus a low amount of  $Ag_2S$  benefits the light absorbance and charge carries transfer. Along with the in-situ formation of metallic Ag at  $Ag_3PO_4$  surfaces, a stable and well-defined  $Ag_2S/Ag/Ag_3PO_4$  Z-scheme system ensures the excellent photocatalytic performance towards



**Fig. 10.** (a) Steady-state PL ( $\lambda_{ex}$  = 320 nm), (b) time-resolved PL spectra, and (c) EIS Nyquist plots for pure Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites.

MO degradation. In the case of NIR light irradiation, the activity enhancement of  $Ag_2S/Ag_3PO_4$  composites mainly arises from the sensitization effect of  $Ag_2S$ . Thus, the more  $Ag_2S$  load on the  $Ag_3PO_4$  surfaces, the more photons generate and the higher photocatalytic activity is. However, with excess  $Ag_2S$  in the  $Ag_2S/Ag_3PO_4$  composites, the well-contact interface between  $Ag_2S$  and  $Ag_3PO_4$  is greatly destroyed. In this case, the electrons generated in  $Ag_2S$  are unfavorable to transfer to  $Ag_3PO_4$  but combine easily with holes at  $Ag_2S$  itself, resulting in the decreased photocatalytic activity therefore.

#### 4. Conclusions

Highly robust Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> core/shell composites with different Ag<sub>2</sub>S contents have been prepared by the simple anion-exchange method. The photocatalytic activity and stability of Ag<sub>3</sub>PO<sub>4</sub> towards

MO degradation under both visible and NIR light are greatly improved by the introduction of Ag<sub>2</sub>S. The enhanced activity of Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> composites is due to the effective separation and fast transfer of photogenerated charge carriers. The photocatalytic mechanism study highlights that the *in-situ* formed metallic Ag, can respectively act as charge transmission bridges and electron traps under visible and NIR light irradiation, and further help to construct the stable Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> and Ag/Ag<sub>2</sub>S/Ag<sub>3</sub>PO<sub>4</sub> systems. This research shows the possibility of utilizing Ag<sub>2</sub>S as photosensitizer or appropriate narrow-band-gap semiconductor to achieve highly efficient composite photocatalysts working under visible and NIR light irradiation.

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#### Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at http://dx.doi.org/10.1016/j.apcatb.2017.03.022.

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